



Lattice dynamics and Raman spectrum of BaZrO₃ single crystals

Constance Toulouse, Danila Amoroso, Cong Xin, Philippe Veber, Monica Ciomaga Hatnean, Geetha Balakrishnan, Mario Maglione, Philippe Ghosez, Jens Kreisel, Mael Guennou

► To cite this version:

Constance Toulouse, Danila Amoroso, Cong Xin, Philippe Veber, Monica Ciomaga Hatnean, et al.. Lattice dynamics and Raman spectrum of BaZrO₃ single crystals. *Physical Review B: Condensed Matter and Materials Physics* (1998-2015), 2019, 100 (13), 134102 (8 p.). 10.1103/PhysRevB.100.134102 . hal-02367817

HAL Id: hal-02367817

<https://univ-lyon1.hal.science/hal-02367817>

Submitted on 30 Sep 2020

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Lattice dynamics and Raman spectrum of BaZrO₃ single crystals

Constance Toulouse ^{1,2} Danila Amoroso,^{3,4,5} Cong Xin,^{2,4} Philippe Veber,^{6,4} Monica Ciomaga Hatnean ⁷,
Geetha Balakrishnan,⁷ Mario Maglione,⁴ Philippe Ghosez,³ Jens Kreisel,¹ and Mael Guennou ¹

¹*Physics and Materials Science Research Unit, University of Luxembourg, 41 Rue du Brill, L-4422 Belvaux, Luxembourg*

²*Materials Research and Technology Department, Luxembourg Institute of Science and Technology, 41 rue du Brill, L-4422 Belvaux, Luxembourg*

³*Theoretical Materials Physics, Quantum Materials Center (Q-MAT), Complex and Entangled Systems from Atoms to Materials (CESAM), University of Liège, Quartier Agora, Allée du Six Aout, B-4000 Liège, Belgique*

⁴*Institut de Chimie de la Matière Condensée de Bordeaux (ICMCB)-UMR 5026, 87, Avenue du Docteur Schweitzer, F-33608 Pessac, France*

⁵*National Research Council CNR-SPIN, c/o Università degli Studi "G. D'Annunzio," I-66100 Chieti, Italy*

⁶*Institut Lumière-Matière (ILM)-UMR 5306, Université Claude Bernard Lyon 1, Campus LyonTech-La Doua, 10 rue Ada Byron, F-69622 Villeurbanne, France*

⁷*Physics Department, University of Warwick, Coventry, CV4 7AL, United Kingdom*



(Received 3 July 2019; published 2 October 2019)

BaZrO₃ is a perovskite that remains in the simple cubic phase at all temperatures, hence with no first-order Raman-active phonon mode allowed by symmetry. Yet, it exhibits an intense Raman spectrum with sharp and well-defined features. Here, we report the evolution of the Raman spectrum of BaZrO₃ single crystals in a broad temperature range (4–1200 K) and discuss its origin with the support of detailed first-principle calculations of the lattice dynamics. Phonon calculations are performed not only for the cubic phase of BaZrO₃, but also for the low-symmetry phases with octahedra tilts that have been suspected to exist at the nanoscale. We show that the Raman spectrum shows no direct evidence for these nanodomains, but can instead be explained by classical second-order Raman scattering. We provide an assignment of the dominant features to phonon mode combinations. In particular, we show that the high frequency range of the spectrum is dominated by overtones and shows an image of the phonon density of states corresponding to the stretching modes of the oxygen octahedra.

DOI: [10.1103/PhysRevB.100.134102](https://doi.org/10.1103/PhysRevB.100.134102)

I. INTRODUCTION

Barium zirconate (BaZrO₃) belongs to the family of chemically simple perovskite oxides. It has a Goldschmidt tolerance factor close to 1, and crystallizes in the ideal cubic structure with the $Pm\bar{3}m$ space group. It remains cubic at ambient conditions and down to liquid helium temperatures [1,2]. As a functional material, BaZrO₃ has attracted attention as a high-temperature proton conductor [3], it can be used in solid oxide fuel cells [4] and sensors [5], or as nanoparticles for increasing the vortex pinning in iron-based superconductor thin films [6]. It is also of interest as one of the end-members of the (Ba, Ca)(Ti, Zr)O₃ system for high-performance lead-free piezoelectrics [7–10]. At a fundamental level, its lattice dynamics and stability were found more complex than anticipated, which has drawn attention to its low-temperature structure and properties. First-principle calculations predict that the simple cubic phase is in fact slightly unstable, and that the ground state is a distorted phase of lower symmetry with octahedra tilts [2,9,11,12]. Yet, a phase transition has never been observed experimentally at ambient pressure at any temperature. However, BaZrO₃ exhibits a structural phase transition under pressure around 11 GPa, presumably toward a tetragonal tilted phase [13]. Such a large stability range of the cubic phase is in fact remarkable, and an exception in simple perovskite oxides. The overwhelming majority of ABO₃ compounds are distorted at ambient conditions and

their cubic phase, when experimentally observable, is only stabilized at high pressures or temperatures, for example, above 1300 K in CaTiO₃ [14] and above 820 K or 14 GPa in LaAlO₃ [15,16].

Distorted perovskites all possess first-order Raman-active modes, some of them directly related to the corresponding order parameters. In contrast, simple cubic perovskites have no Raman-active phonon mode so that no first-order Raman spectrum is allowed by symmetry. In spite of this, past studies of BaZrO₃ have revealed an intense Raman spectrum [1,13,17–19] with relatively sharp features. Similar situations have been encountered in some other cubic perovskites. The most well-known case is the incipient ferroelectric SrTiO₃ whose Raman spectrum has been studied in detail [20–24]. Other cases include KTaO₃ [21], BaSnO₃ [25], or even BaTiO₃ in its cubic phase above T_c [26]. In general, the interpretation and understanding of Raman spectra in cubic perovskites is not straightforward, and the assignment of Raman bands to precise structural or vibrational features requires more extensive experimental and theoretical work.

Two types of interpretations have been put forward in past studies to explain Raman spectra of simple cubic perovskites. The first explains the spectra by second-order Raman processes, where two phonons with opposite wave-vectors are involved in the same scattering event. Second-order scattering is always symmetry-allowed; it is expected to be weaker in intensity, but, in the absence of first-order scattering, it may

become the dominant process. Conservation of momentum for second-order processes no longer restricts the phonons involved to the center of the Brillouin zone, and selection rules then allow for a large number of possible combinations. An accurate assignment of Raman features to specific vibrations or combination of vibrations then requires a comprehensive knowledge of the whole phonon dispersion, which may come from experiment or lattice dynamics calculations—as shown by the long history of SrTiO_3 second-order Raman interpretations [20–23]. Salient features in the Raman spectrum are typically assigned to combinations of zone-boundary phonon modes where singularities in the density of states are expected.

The second interpretation invokes some local symmetry breaking that relaxes the Raman selection rules and allows for Raman scattering to occur, while averaging to a cubic symmetry at the macroscopic scale probed by X-ray diffraction. This is typically the case in BaTiO_3 where Ti-displacements remain correlated above T_c in the macroscopically cubic phase, which results in the persistence of Raman peaks, but also deviations in other physical properties, like birefringence [27]. In this case, the Raman scattering process is rather first-order in nature. This explanation has been proposed for BaZrO_3 also [13,19,28], whereby the local distortions could be caused by octahedra tilts, and related to the structural instability revealed by first-principle calculations. However, the detailed interpretation remains an open problem. This has largely to do with the fact that, experimentally, BaZrO_3 single crystals are difficult to grow because of their high melting point. As a result, it has been usually studied as polycrystalline samples [18,19], and comparatively less than other perovskites.

In this work, we use recently synthesized BaZrO_3 single crystals [1] to investigate the lattice dynamics by a combination of single crystal Raman spectroscopy in a broad range of temperatures at ambient pressure, density functional theory (DFT)-based calculations of the dispersion and mode symmetry analysis. We demonstrate that the Raman spectrum of BaZrO_3 is dominated by second-order Raman scattering, without any evidence for octahedra tilts at the local scale. We also provide assignments of the most intense modes of the Raman spectra in terms of vibration pattern.

II. METHODS AND EXPERIMENTAL DETAILS

The single crystals used in this work have been described in Ref. [1]. They include a crystal grown by the floating zone technique and two crystals purchased from Crystal Base Co. Ltd. and grown by the tri-arc method [1,29]. No significant difference has been observed in the Raman spectra of the different crystals. Polarized Raman spectra were recorded in a back-scattering geometry with a Renishaw inVia Reflex Raman Microscope using the 442-nm laser line of a He-Cd laser with an output power of 58 mW. Low temperature measurements have been performed using an Oxford Microstat-Hire open-cycle He cryostat together with an Oxford Mercury ITC controller. High temperature measurements have been recorded inside a TS1500 Linkam stage.

First-principle calculations of structural and dynamical properties rely on DFT and density functional perturbation theory (DFPT) [30], as implemented in the ABINIT package

[31,32]. The exchange-correlation potential was evaluated within the generalized gradient approximation (GGA) using the Wu-Cohen (WC) functional [33]. Norm-conserving pseudopotentials [34] have been employed with the following orbitals considered as the valence states: $5s$, $5p$, and $6s$ for Ba $4s$, $4p$, $4d$, and $5s$ for Zr, and $2s$ and $2p$ for O. The energy cutoff for the expansion of the electronic wave functions has been fixed to 45 Ha and we used a $6 \times 6 \times 6$ k-point mesh for the Brillouin zone sampling. Phonon calculations have been performed on the fully relaxed cubic and antiferrodistortive (AFD) structures, for which the structural optimization reached values of below $\sim 10^{-4}$ eV/Å. The q points for the phonon dispersion curves of the 5-atom cubic BZO included Γ , X, M, R, and the Δ point halfway from Γ to R of the simple cubic Brillouin zone.

Symmetry analysis of the phonon modes was done with the help of programs from the Bilbao crystallographic server [35,36] and the ISOTROPY software suite [37].

III. RESULTS

A. Lattice dynamics calculations

We first computed the lattice dynamics for the $Pm\bar{3}m$ cubic phase over the full Brillouin zone. The dispersion relations together with the symmetry assignment for the modes and the branches is given in Fig. 1. This is essentially in agreement with previous reports of similar calculations [2,12]. We find that the cubic phase is dynamically unstable, with an unstable phonon mode at the R point of the Brillouin zone corresponding to the tilts of the ZrO_6 octahedra.

There are few experimental studies on the lattice dynamics of cubic BaZrO_3 that can be used for comparison. The polar modes have been measured by IR spectroscopy on polycrystalline samples [38]. The three polar modes in cubic perovskites are commonly referred to as “Slater” mode (anti-phase motion of the B cation with respect to the oxygen octahedra), “Axe” mode (anti-phase motion of the apical oxygens with respect to the other oxygen atoms of the octahedra) and “Last” mode (anti-phase motion of the A cation and the ZrO_6 octahedra) as described in relation to ferroelectric transitions [39]. Their DFT-computed frequencies are in agreement with experimental IR values from Ref. [38] within 20 cm^{-1} (our calculated frequencies of the TO modes are 96, 193 and 503 cm^{-1} , respectively, for 116, 214, and 520 cm^{-1} reported in Ref. [38]). The LO frequencies show a better agreement, with calculated (respectively experimental) values at 133 (141), 366 (380), and $692 (692) \text{ cm}^{-1}$.

We then calculated the lattice dynamics for the phases resulting from the condensation of the unstable R-point phonon mode, namely the tetragonal $I4/mcm$, orthorhombic $Imma$ and rhombohedral $R\bar{3}c$ phases corresponding to out-of-phase oxygen rotations around one, two and three directions of the pseudo-cubic reference cell, respectively. All three AFD phases are dynamically stable, as revealed by the calculation of the phonon dispersion curves (see Fig. 2 in the Supplemental Material [40]), and lower in energy than the cubic phase. Maximum energy gain is obtained with octahedra rotation around the [001] direction in the tetragonal phase, which also displays the largest

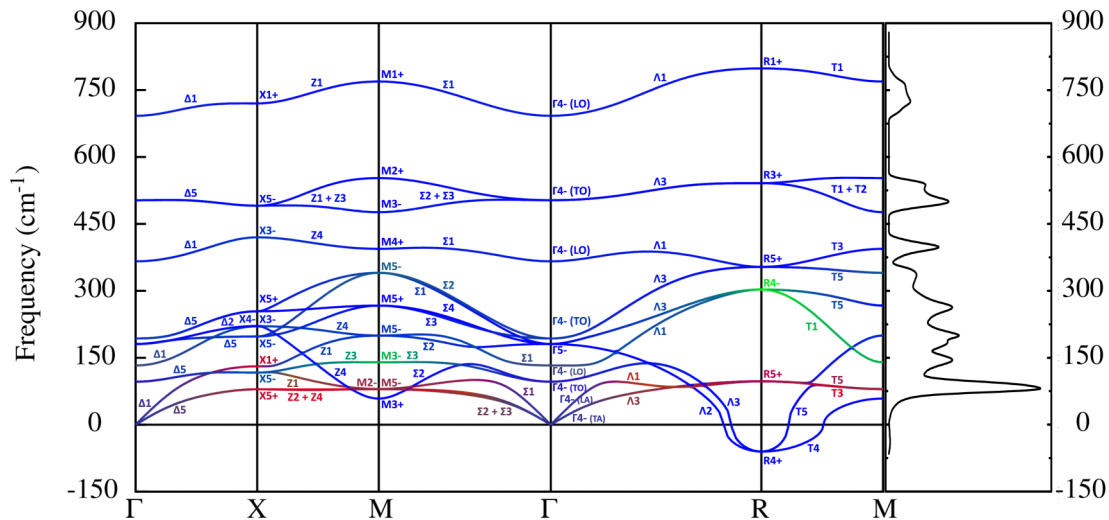


FIG. 1. Dispersion curves of the phonon modes of BaZrO₃ in its cubic phase computed by DFT. Negative frequency values refer to imaginary frequencies. Colors correspond to the atoms involved in the vibration: blue for Oxygen, green for Zirconium, and red for Barium atoms. The total density of states (DOS) is plotted on the right.

distortion, as shown in Fig. 1 of the Supplementary Material [40]. Nevertheless, those phases can be considered quasi-degenerate, the difference in energy being within 0.3 meV/f.u. [9].

Details of the different phonon modes, along with their symmetry assignments, for all four phases and their correspondences between phases is shown on Table I. For each phase, we identified the predicted Raman active modes and their relation to the modes in the parent phase. In all cases, the Raman-active modes originate from modes at the R point folded onto the Γ point. In principle, Raman spectroscopy can distinguish between these structural variants because they have different numbers of Raman modes with different degeneracies and symmetries: The tetragonal phase has 7 Raman modes ($A_{1g} + B_{1g} + 2B_{2g} + 3E_g$), the orthorhombic phase has 12 ($3A_g + 2B_{1g} + 3B_{2g} + 4B_{3g}$), and the rhombohedral phase has 5 ($A_{1g} + 4E_g$). However, in practice, we observe that these modes originate in most cases from the same degenerate vibrations of the cubic phase, and the very small distortion makes the mode splitting extremely small, sometimes below the resolution limit of classical instruments. Finally, at the level of the phonon DOS, the three phases look almost the same [40]. The notable exceptions are: (i) the octahedra breathing mode, which is only Raman active in the orthorhombic phase, and (ii) the soft tilt modes, whose frequencies vary greatly depending on the tilt system. This, combined to the fact that well-oriented single domains are difficult to obtain when going through a ferroelastic transition, makes it non-trivial to identify which phase is actually stabilized.

In addition, to investigate the hypothesis of a second-order Raman spectrum, we determined the symmetries of the two-phonon states for all phonons at the zone center and at the high-symmetry points at the zone boundary (R , X and M) where peaks and kinks in the second-order spectrum are expected due to van Hove singularities [41]. For a cubic perovskite with point group $m\bar{3}m$, the problem is in principle similar to the cases of silicon, diamond, or cubic SiC [42–44],

only with many more possible phonon combinations. The point group $m\bar{3}m$ has three Raman-active representations A_{1g} , E_g , and T_{2g} . Tables providing the symmetry of combinations for special k points and relevant phonon symmetries in BaZrO_3 are provided in the Supplemental Material [40]. An important and general result is that overtones are always Raman-active. Besides, the majority of combinations contain more than one Raman-active representation, so that, generally speaking, very few of them will exhibit clear selection rules.

B. Temperature dependent Raman spectroscopy

The polarized Raman spectra of our BaZrO₃ single crystals taken from 4 K to room temperature and up to 1200 K are shown on Figs. 2(a) and 2(b), respectively.

The first result stemming from these spectra is that the behavior of the spectra between 4 and 1200 K shows no evidence of a structural transition occurring in BaZrO₃ in that temperature range, confirming the stability of the cubic structure.

At low temperatures, approximately below 100 K, a large number of weak thin peaks appear, especially at high energies. Those peaks do not appear in our complementary measurements performed with a different excitation wavelength (532 nm), and we therefore conclude that they are not Raman peaks but rather originate from some luminescence processes. Their detailed assignment is beyond the scope of this work, but we note that they are very similar in positions, shape, and temperature behavior to the lines observed in the Raman spectra of LaAlO_3 , which have been assigned to various intrinsic defects [45].

While the spectrum consists mostly of broad features (reminiscent of second-order scattering), the main dominant peaks are quite sharp, and do not broaden significantly when the temperature increases: their full-width at half maximum (FWHM) is around 20–40 cm^{-1} at 4 K and 40–60 at room temperature [Fig. 2(a)], which is comparable to first-order phonon modes. Surprisingly, the frequencies of these peaks

TABLE I. Optical phonon mode frequencies and symmetries in the different structural phases calculated by DFT (acoustic modes are omitted). The different modes are grouped together based on the symmetry transformations between phases and the modes indicated in bold font correspond to the Raman active modes (for first-order processes).

Cubic ($Pm\bar{3}m$ -221)		Related atomic displacements	Tetra ($I4/m\bar{c}m$ -140)		Ortho ($Imma$ -74)		Rhombo ($R\bar{3}c$ -167)	
TO/LO modes								
-60	R_4^+	oxygen octahedra rotations	23	E_g	20	B_{2g}	73	E_g
			93	A_{1g}	87	B_{1g}	88.5	A_{1g}
97	R_5^+	antiparallel Ba motion	102	E_g	100	B_{2g}	104	E_g
			104	B_{2g}	107	B_{3g}		
96/133	$\Gamma_4^- (T_{1u})$	zone center “Last” modes: Ba displacement against ZrO ₆ octahedra	107	E_u	104	B_{3u}	98	A_{2g}
					107	B_{2u}	112	E_u
181	$\Gamma_5^- (T_{2u})$	silent zone-center mode: antiparallel out-of-plane displace- ment of in-plane oxygens	113	A_{2u}	114	B_{1u}	99	A_{2u}
			192	E_u	186	B_{3u}	188	E_u
193/366	$\Gamma_4^- (T_{1u})$	zone center “Slater” modes: out-of-plane displacements of Zr against in-plane O	195	A_{2u}	194	B_{1u}	193	A_{2u}
			195	E_u	194	A_u	195	E_u
303	R_4^-	Zr antiparallel displacements	200	B_{1u}	204	B_{3u}	204	A_{1u}
			301	A_{1u}	301	B_{1u}	301	A_{1u}
354	R_5^+	oxygen octahedra shearing modes	302	E_u	302	B_{1u}	303	E_u
					302	B_{2u}		
503/692	$\Gamma_4^- (T_{1u})$	oxygen octahedra shearing modes	356	E_g	354	B_{2g}	358	E_g
			358	B_{2g}	356	B_{3g}		
503/692	$\Gamma_4^- (T_{1u})$	zone center “Axe” modes: displacement of apical oxygens against in-plane oxygens	360	A_g	353	A_g	353	A_{2g}
			494	E_u	494	B_{3u}	497	E_u
541	R_3^+	Jahn-Teller-like distortions of oxygen octahedra	495	B_{2u}	495	B_{2u}		
			496	A_{2u}	497	B_{1u}	494	A_{2u}
799	R_1^+	oxygen octahedra breathing mode	534	A_{2g}	535	B_{1g}	535	E_g
			535	B_{1g}	535	B_{3g}		
799	R_1^+	oxygen octahedra breathing mode	791	A_{2g}	791	B_{3g}	792	A_{2g}

are shifting much less with temperature than commonly observed for thermal softening of phonon modes. As an example, the three dominant peaks that are sharper and more intense situated at 648, 741, and 838 cm^{-1} at 4 K are found at 1200 K at 613, 728, and 843 cm^{-1} , respectively, corresponding to relative shifts of -4.5 , -1.5 , and $+0.5 \times 10^{-5} \text{ K}^{-1}$.

To separate the spectrum into its different symmetry components, we performed polarized Raman measurements. Choosing x , y , and z along the cubic crystallographic axes, and labeling x' and y' axes rotated by 45° along the z axis, the polarization components can be accessed in the following backscattering geometries (in Porto notation):

$$z(xx)z \rightarrow A_{1g} + E_g,$$

$$z(xy)z \rightarrow T_{2g},$$

$$z(x'x')z \rightarrow A_{1g} + T_{2g},$$

$$z(x'y')z \rightarrow E_g.$$

The spectra recorded in those four configurations at 4 K are shown in Fig. 2(c). The E_g and T_{2g} spectra are found to be weaker. The broad peak at 450 cm^{-1} appears more strongly in the T_{2g} spectrum, but no other feature can be conclusively assigned to these symmetries only. As a result, we will

consider in the following that the spectrum has essentially the full symmetry A_{1g} .

IV. DISCUSSION: INTERPRETATION/ORIGIN OF THE RAMAN SPECTRUM

A. High-frequency part of the spectrum

To discuss the interpretation of the Raman spectrum of BaZrO₃ and try to understand its origin, we first focus on the high-frequency part of the spectrum (above 800 cm^{-1}). This range is above the (one-)phonon frequency range, so that the assignment of those features to second-order scattering with the creation (or annihilation) of two phonons is in fact the only possible. Since overtones are always Raman-allowed, we start by comparing in Fig. 3 the overtone density of states with the experimental spectra taken at 4 K. The agreement is very good, and the experimental spectrum looks like a faithful image of the overtone DOS. The phonon modes involved are the high energy modes corresponding to the octahedra stretching branches. There is only a small shift in frequency between the experimental spectrum and the DOS, most likely explained by the usual mismatch in volume found in DFT calculations.

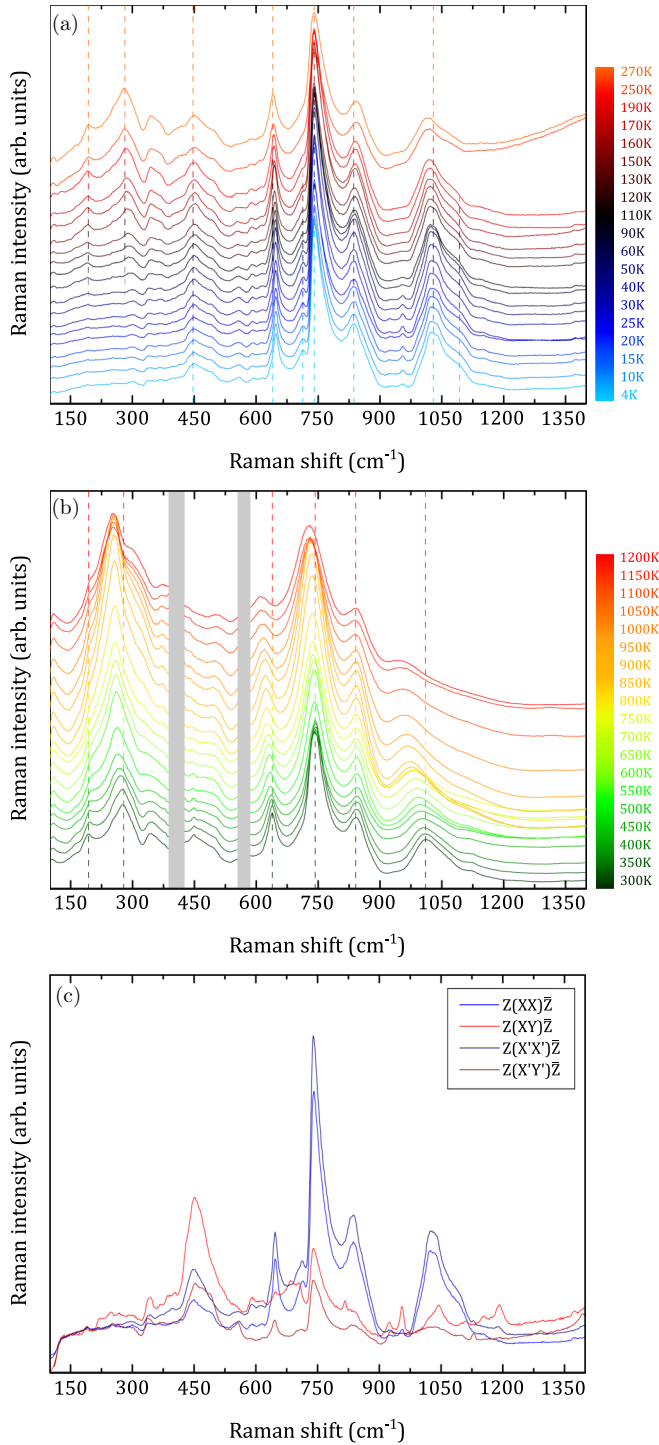


FIG. 2. Experimental Raman data showing (a) the low-temperature and (b) high-temperature dependences of the Raman spectra measured in parallel polarization configuration (the crossed polarization spectra are shown in the Supplemental Material [40]), the dotted lines are guidelines to follow the more intense modes that are discussed further on, and (c) the polarized Raman measurements at 4 K, in the 4 polarization configurations described in the text.

Assignment of the kinks in the DOS/experimental spectrum can be done based on the shape of the spectrum. Of interest is the main peak at 1028 cm⁻¹ which can be assigned to the overtone of the IR-active TO mode and the 1090 cm⁻¹

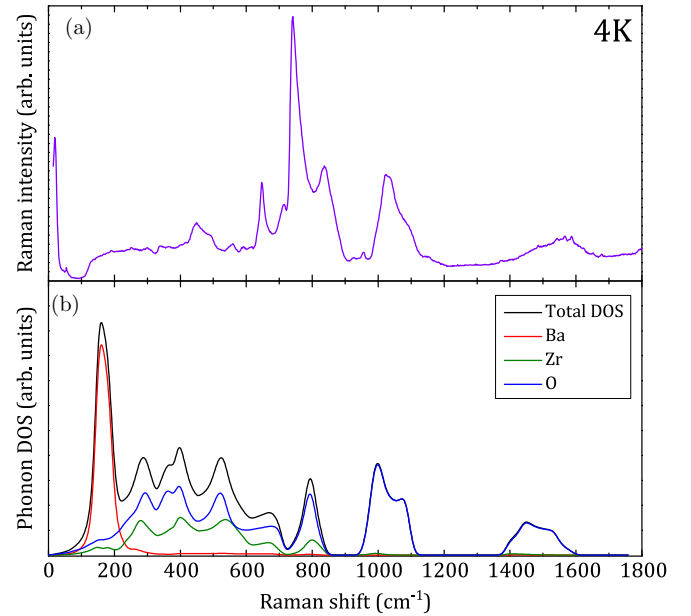


FIG. 3. (a) Raman spectrum of BaZrO₃ at 4 K, in the z(xx)z geometry. (b) Calculated overtone density of states, whereby the respective contributions of different atoms are separated and shown alongside the total density of states.

to an overtone of the R_3^+ mode. The second kink around 1500 cm⁻¹ corresponds to the oxygen octahedra breathing mode, in the highest branch. Interestingly, we can notice that zone-boundary points account for the kinks in the spectrum, but do not give rise to increased intensity. Also, the R point, from which the structural instability arises, does not play any particular role in the understanding of the high-frequency part of the Raman spectrum.

B. Low-frequency spectrum

For the lower part of the spectrum (below 800 cm⁻¹), the overtone density of states clearly does not match the observed spectrum. In particular, it does not show any obvious assignment for the three dominant features of the experimental spectrum at 451, 648, and 741 cm⁻¹, and conversely, the experimental spectrum does not show any sign of the intense peak in the density of states associated to Ba motion. We discuss their possible origins below.

Let us first consider the hypothesis of first-order scattering allowed by a local symmetry breaking, with the idea that the structure at the local scale is close to the tilted phases described previously. Within that model, we expect the most intense features of the Raman spectrum to correspond to frequencies of phonon modes that become Raman active in the tilted phases. However, the comparison of the frequencies of the phonon modes computed by DFT in the three possible distorted phase given in Table I does not support this hypothesis. The three most intense experimental Raman modes at 741, 648, and 451 cm⁻¹ are not found in the DFT modes frequencies of any of the structural phases proposed. Therefore, the scenario involving first-order Raman signal stemming from nanodomains of one of the three possible distorted structures does not provide a convincing explanation

TABLE II. Experimental frequencies of the intense Raman peaks and their tentative attribution to combinations of phonons calculated by DFT. The comparison with the experimental infrared frequencies derived from [38] is also shown. The Raman frequencies are given at 4 K, unless specified otherwise. Overtones are explicitly indicated while \oplus indicates a combination of different phonon modes with identical symmetries.

Experiment		DFT theoretical assignment—second order		Literature (exp.)
freq. (cm ⁻¹)	freq. (cm ⁻¹)	Combination	Associated atomic displacement	Nuzhnyy <i>et al.</i> [38]
451	451	$R_5^+ \oplus R_5^+$	Ba (antiparallel) + octahedra rotations	
648	636	$\Gamma_4^- \oplus \Gamma_4^-$	Last(LO) + Axe(TO) zone center modes	661
713	707	overtone R_5^+	“scissoring” mode of octahedra	
741	732	overtone Γ_4^-	Slater(LO) zone center mode	760
	745	$X_5^- \oplus X_5^-$	in-plane antiparallel displacement of apical O + antiparallel apical oxygen displacements	
838	840	overtone X_3^-	displacements of Zr against in-plane oxygens Last (LO) + Axe (LO)	833
1028	1006	overtone Γ_4^-	Zone center Axe(TO) mode	1040
1045	1059	$\Gamma_4^- \oplus \Gamma_4^-$	Zone center Axe(LO) + Slater(LO) modes	1072
1090	1083	overtone R_3^+	Jahn-Teller-like octahedra distortions	
1573	1538	overtone M_1^+	oxygen octahedra breathing mode (x,y)	
	1597	overtone R_1^+	oxygen octahedra breathing mode	
193 (Room T)	193	overtone Γ_4^-	Last(TO) zone center mode	232
	195	overtone R_5^+	antiparallel Ba displacements	
277 (Room T)	277	$\Gamma_4^- \oplus \Gamma_5^-$	Last(TO) + shear mode octahedra	
	277	$X_5^+ \oplus X_5^+$	antiparallel Ba displacements (x,y) + antiparallel oxygens (x,y) displacements	
	280	overtone M_3^-	antiparallel Zr displacements (z)	
253 (1200 K)	266	overtone Γ_4^-	Last(LO) zone center mode	282
	262	overtone X_1^+	antiparallel Ba displacements (x)	

of our experimental phonon modes, unless one assumes other type of distortions.

Another hypothesis is to assign the peaks to the polar modes that are activated by disorder. This was for example proposed for the intense modes in BaSnO₃ that bears similarities with the BaZrO₃ peaks [25]. We investigated that hypothesis by taking the IR experimental values of the TO and LO frequencies, extracted from Refs. [12,38] and comparing them to our experimental data. The frequencies do not match our data either (cf. Table. III in the Supplemental Material [40]). We therefore discard this hypothesis. More generally, we observe that there is a large gap in the phonon density of states between 553 and 692 cm⁻¹, so that it is nearly impossible to assign the peak at 648 cm⁻¹ to any first-order phonon scattering process.

We therefore conclude that the low-frequency spectrum is also dominantly of second-order character. We sought possible assignments among the symmetry-allowed overtones and combinations of phonon frequencies at high-symmetry points at the zone center and the zone boundary, as described in Sec. III A. We kept the modes combinations with the closest frequencies to our experimental frequencies. These assignments are shown in Table II. Interestingly, the two dominant peaks at 741 and 648 cm⁻¹ are assigned to combinations of polar modes at the zone center. In some cases, more than one combination can match the observed frequency. It must also be pointed out that assignment at low frequencies must be taken with care for two reasons. First, the real frequency of the soft mode in the cubic phase remains unknown. Consid-

ering its energy in the stable tilted phases (Table I), we can reasonably hypothesize that it is below 100 cm⁻¹, and unlikely to affect our interpretation of the high frequency spectrum. Second, the low-frequency part may contain subtraction combinations that we are not considered here.

In addition, our assignments are based on the concordance between experimental and theoretical frequencies, which arguably comes with some uncertainty. In Ref. [38], the discrepancy between experimental infrared and theoretical frequencies for the polar modes can go up to 30 cm⁻¹. In this work, the misfit between the experimental and theoretical maxima of the intense band around 1000 cm⁻¹ is just over 20 cm⁻¹, yielding an uncertainty of around 10 cm⁻¹ for the phonon frequencies calculated by DFT. Because of that, we decided to take a error margin of about 20 cm⁻¹ when searching for theoretical modes matching our Raman data.

The general temperature dependence of the spectrum is not conclusive in itself but is consistent with a second-order assignment. The temperature dependence of Raman intensity can in principle allows to distinguish first- from second-order scattering, because both have different dependences on the Bose-Einstein occupation factor n : $(1+n)$ for first order, $(1+n)(2+n)$ for overtones, and $(n+1)^2$ for combinations of different phonons [41]. Qualitatively, this leads to an enhanced intensity at low frequencies and high temperatures, which is in agreement with what is observed in Fig. 2(b). On the other hand, the Raman intensity can be expected to decrease with temperature, due to thermal expansion [28], with an unknown rate, which makes the analysis difficult in

practice. The temperature dependence of the peak positions is found to be equally inconclusive: they show relative shifts of -4.5 and $-1.5 \times 10^{-5} \text{ cm}^{-1} \text{ K}^{-1}$, respectively, for the main peaks. Taking a literature value for the volumic thermal expansion of BaZrO_3 : $2.1 \times 10^{-5} \text{ cm}^{-1} \text{ K}^{-1}$ [46,47], we would obtain Grüneisen parameters of 2.1 and 0.7, respectively, which can support both the first and second-order nature of the peaks.

Finally, we note that the pressure dependence of the Raman modes can be also ambiguous to distinguish between first- from second-order processes. With increasing pressure, the room-temperature Raman spectrum of BaZrO_3 tends to vanish, which was taken as a support for a scenario where locally distortions give space to macroscopically cubic phase [13]. However, a classical second-order spectrum is also expected to decrease in intensity with pressure [48]—as it was observed in SrTiO_3 [49] and CaTiO_3 [50], so that this observation alone can support both hypotheses.

C. Relation with local disorder

In summary, we find that the whole Raman spectrum of BaZrO_3 can be explained by second-order Raman scattering. We do not find that the phonons at the *R*-point, where BaZrO_3 has its structural instability, play a special role. In particular, we find no evidence for scattering directly related to octahedra tilts, and no direct support for the hypothesis of distorted nanodomains.

However, it is still very likely that the overall intensity of the scattering finds its origin in another type of local disorder. Let us first point out that the presence of a strong Raman spectrum in cubic perovskites is not a general property. This can be appreciated by comparing the relative strengths of second-versus first-order spectra: typical perovskites such as LaAlO_3 or PbTiO_3 exhibit in their cubic phase a Raman spectrum that is extremely weak compared to the intensity of the first-order Raman modes allowed in their low-symmetry phase. In contrast, in SrTiO_3 or CaTiO_3 , both contributions are of equal strength. Those compounds also share common experimental signatures related to local disorder of the Ti^{4+} ion in the octahedral cage, with the presence of intense diffuse scattering [51] and distinctive features in XANES [52]. We therefore expect that studies of the local structure in BaZrO_3 will be insightful to reveal the details and dynamics of disorder, and we

note that a previous study has already proposed anomalous Ba displacements, revealed by anomalously large Debye-Waller (DW) factor in BaZrO_3 structural refinements [11].

V. CONCLUSION

In summary, we have provided measurements of the Raman spectrum of BaZrO_3 single crystals in a wide temperature range, and analyzed it in the light of comprehensive lattice dynamical calculations of BaZrO_3 in its cubic phase as well as in its distorted phases with octahedra tilts. We clearly identified a major contribution from second-order Raman scattering, especially overtones of the octahedra breathing modes at high frequencies. Other salient features of the Raman spectrum have also been tentatively assigned to overtones and mode combinations. Raman spectroscopy therefore does not directly support the hypothesis of locally tilted nanodomains, although local disorder is still believed to play a role in the general intensity of the Raman scattering. In addition, we have attracted attention to the expected similarities between the Raman spectrum of all tilted phases, which calls for some care in the claims for transition to orthorhombic phases in high-pressure measurements of BaZrO_3 and SrTiO_3 . Finally, the low-frequency lattice dynamics at the zone-boundary and its behavior with temperature remains to be experimentally determined and will be the object of further work.

ACKNOWLEDGMENTS

This work was supported by the Innovative Training Networks (ITN) Marie Skłodowska-Curie Actions-European Joint Doctorate in Functional Material Research (EJD-FunMat) (Project No. 641640). DFT-based calculations have been performed on the NIC4 cluster hosted at the University of Liège, within the ‘Consortium des Équipements de Calcul Intensif’ (CÉCI), funded by F.R.S-FNRS (Grant No. 2.5020.1) and by the Walloon Region. C.T., M.G., J.K. acknowledge financial support from the Fond National de Recherche Luxembourg through a PEARL Grant (No. FNR/P12/4853155/Kreisel). The work at the University of Warwick was supported by the EPSRC, UK (Grant No. EP/M028771/1).

-
- [1] C. Xin, P. Veber, M. Guennou, C. Toulouse, N. Valle, M. C. Hatnean, G. Balakrishnan, R. Haumont, R. S. Martin, M. Velazquez *et al.*, *CrystEngComm* **21**, 502 (2019).
 - [2] A. R. Akbarzadeh, I. Kornev, C. Malibert, L. Bellaiche, and J. M. Kiat, *Phys. Rev. B* **72**, 205104 (2005).
 - [3] K. Kreuer, *Annu. Rev. Mater. Res.* **33**, 333 (2003).
 - [4] D. Pergolesi, E. Fabbri, A. D’Epifanio, E. Di Bartolomeo, A. Tebano, S. Sanna, S. Licocchia, G. Balestrino, and E. Traversa, *Nat. Mater.* **9**, 846 (2010).
 - [5] X. Chen, L. Rieth, M. S. Miller, and F. Solzbacher, *Sensors Actuators B: Chem.* **137**, 578 (2009).
 - [6] M. Miura, B. Maiorov, T. Kato, T. Shimode, K. Wada, S. Adachi, and K. Tanabe, *Nat. Commun.* **4**, 2499 (2013).
 - [7] D. S. Keeble, F. Benabdallah, P. A. Thomas, M. Maglione, and J. Kreisel, *Appl. Phys. Lett.* **102**, 092903 (2013).
 - [8] W. Liu and X. Ren, *Phys. Rev. Lett.* **103**, 257602 (2009).
 - [9] D. Amoroso, A. Cano, and P. Ghosez, *Phys. Rev. B* **97**, 174108 (2018).
 - [10] G. Buse, C. Xin, P. Marchet, A. Borta-Boyon, M. Pham-Thi, H. Cabane, E. Veron, M. Josse, M. Velazquez, M. Lahaye *et al.*, *Crystal Growth Design* **18**, 5874 (2018).

- [11] A. I. Lebedev and I. A. Sluchinskaya, *Phys. Solid State* **55**, 1941 (2013).
- [12] J. W. Bennett, I. Grinberg, and A. M. Rappe, *Phys. Rev. B* **73**, 180102(R) (2006).
- [13] C. Chemarin, N. Rosman, T. Pagnier, and G. Lucazeau, *J. Solid State Chem.* **149**, 298 (2000).
- [14] S. A. T. Redfern, *J. Phys.: Condens. Matter* **8**, 8267 (1996).
- [15] S. A. Hayward, F. D. Morrison, S. A. T. Redfern, E. K. H. Salje, J. F. Scott, K. S. Knight, S. Tarantino, A. M. Glazer, V. Shuvaeva, P. Daniel *et al.*, *Phys. Rev. B* **72**, 054110 (2005).
- [16] M. Guennou, P. Bouvier, G. Garbarino, and J. Kreisel, *J. Phys.: Condens. Matter* **23**, 395401 (2011).
- [17] M. A. Helal, T. Mori, and S. Kojima, *Ferroelectrics* **499**, 107 (2016).
- [18] I. Charrier-Cougoulic, T. Pagnier, and G. Lucazeau, *J. Solid State Chem.* **142**, 220 (1999).
- [19] P. Colomban and A. Slodczyk, *Acta Phys. Pol. A* **116**, 7 (2009).
- [20] P. A. Fleury, J. F. Scott, and J. M. Worlock, *Phys. Rev. Lett.* **21**, 16 (1968).
- [21] W. G. Nilsen and J. G. Skinner, *J. Chem. Phys.* **48**, 2240 (1968).
- [22] C. Perry, J. Fertel, and T. McNelly, *J. Chem. Phys.* **47**, 1619 (1967).
- [23] W. G. Stirling, *J. Phys. C* **5**, 2711 (1972).
- [24] A. Grzechnik, G. H. Wolf, and P. F. McMillan, *J. Raman Spectrosc.* **28**, 885 (1997).
- [25] T. N. Stanislavchuk, A. A. Sirenko, A. P. Litvinchuk, X. Luo, and S.-W. Cheong, *J. Appl. Phys.* **112**, 044108 (2012).
- [26] G. Burns and F. H. Dacol, *Phys. Rev. B* **18**, 5750 (1978).
- [27] A. Ziębińska, D. Rytz, K. Szot, M. Górny, and K. Roleder, *J. Phys.: Condens. Matter* **20**, 142202 (2008).
- [28] G. Lucazeau, *J. Raman Spectrosc.* **34**, 478 (2003).
- [29] T. B. Reed and E. R. Pollard, *J. Cryst. Growth* **2**, 243 (1968).
- [30] X. Gonze and C. Lee, *Phys. Rev. B* **55**, 10355 (1997).
- [31] X. Gonze, J.-M. Beuken, R. Caracas, F. Detraux, M. Fuchs, G.-M. Rignanese, L. Sindic, M. Verstraete, G. Zerah, F. Jollet *et al.*, *Comput. Mater. Sci.* **25**, 478 (2002).
- [32] X. Gonze, B. Amadon, P.-M. Anglade, J.-M. Beuken, F. Bottin, P. Boulanger, F. Bruneval, D. Caliste, R. Caraca, M. Côté *et al.*, *Computer Phys. Comm.* **180**, 2582 (2009).
- [33] Z. Wu and R. E. Cohen, *Phys. Rev. B* **73**, 235116 (2006).
- [34] D. R. Hamann, *Phys. Rev. B* **88**, 085117 (2013).
- [35] E. Kroumova, M. Aroyo, J. Perez-Mato, A. Kirov, C. Capillas, S. Ivantchev, and H. Wondratschek, *Phase Trans.* **76**, 155 (2003).
- [36] L. Elcoro, B. Bradlyn, Z. Wang, M. G. Vergniory, J. Cano, C. Felser, B. A. Bernevig, D. Orobengoa, G. de la Flor, and M. I. Aroyo, *J. Appl. Crystallogr.* **50**, 1457 (2017).
- [37] H. T. Stokes, D. M. Hatch, and B. J. Campbell, *ISOTROPY Software Suite*, iso.byu.edu.
- [38] D. Nuzhnyy, J. Petzelt, M. Savinov, T. Ostapchuk, V. Bovtun, M. Kempa, J. Hlinka, V. Buscaglia, M. T. Buscaglia, and P. Nanni, *Phys. Rev. B* **86**, 014106 (2012).
- [39] J. Hlinka, J. Petzelt, S. Kamba, D. Noujni, and T. Ostapchuk, *Phase Trans.* **79**, 41 (2006).
- [40] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevB.100.134102> for Raman spectra, DFT data in all phases and symmetry tables for Raman active combinations.
- [41] M. Cardona, R. K. Chang, G. Güntherodt, M. B. Long, and H. Vogt, *Light Scattering in Solids II: Basic Concepts and Instrumentation*, Topics in Applied Physics (Springer-Verlag, Berlin, 1982).
- [42] P. A. Temple and C. E. Hathaway, *Phys. Rev. B* **7**, 3685 (1973).
- [43] W. Windl, P. Pavone, K. Karch, O. Schütt, D. Strauch, P. Giannozzi, and S. Baroni, *Phys. Rev. B* **48**, 3164 (1993).
- [44] W. Windl, K. Karch, P. Pavone, O. Schütt, D. Strauch, W. H. Weber, K. C. Hass, and L. Rimai, *Phys. Rev. B* **49**, 8764 (1994).
- [45] S. Saha, B.-C. Cao, M. Motapothula, C.-X. Cong, T. Sarkar, A. Srivastava, S. Sarkar, A. Patra, S. Ghosh, Ariando *et al.*, *Sci. Rep.* **6**, 36859 (2016).
- [46] Y. Zhao and D. J. Weidner, *Phys. Chem. Miner.* **18**, 294 (1991).
- [47] S. Yamanaka, M. Fujikane, T. Hamaguchi, H. Muta, T. Oyama, T. Matsuda, S. ichi Kobayashi, and K. Kurosaki, *J. Alloys Comp.* **359**, 109 (2003).
- [48] C. Trallero-Giner and K. Syassen, *Phys. Status Solidi B* **247**, 182 (2010).
- [49] M. Guennou, P. Bouvier, J. Kreisel, and D. Machon, *Phys. Rev. B* **81**, 054115 (2010).
- [50] M. Guennou, P. Bouvier, B. Krikler, J. Kreisel, R. Haumont, and G. Garbarino, *Phys. Rev. B* **82**, 134101 (2010).
- [51] M. Kopecký, J. Fábry, and J. Kub, *J. Appl. Cryst.* **45**, 393 (2019).
- [52] J. P. Itié, B. Couzinet, A. C. Dhaussy, A. M. Flank, N. Jaouen, P. Lagarde, and A. Polian, *High Press. Res.* **26**, 325 (2006).